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OVERYOLTAGE LOWERING THE SUPERIAMSTON IN HYDROGEN

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A great number of studies has been devoted to investigating possibilities of changing the discharge speed constant in hydrogen, and consequently, the possibility of changing the interior of its overvoltage supertension during electrolysis. Among these studies, the most interesting from the applied point of view are those dealing with the effect of the material and structure of the electrode surface on the kinetics of the electrode process. Thus, for instance, Stender and Pecharskaya [(1)], studying the separation of hydrogen in the presence of technological current density on a series of metals and alloys, established the possibility of a sharp drop in the supertension by applying tungsten and tungsten-nickel cathodes. Positive results from the use of alloys were noted also by other authors. Kabanov and Mosentsveyg [2] discovered the relation between hydrogen supertension and the content of fixed oxygen on the surface of iron electrodes. And finally, the research on hydrogen supertension on metal electrodes with a highly developed surfaces (metalloceramics and powders) conducted by Kuzmin [3] (iron), Murtazaev [4] (cobalt) and Maytak [5] (copper), indicate great possibilities of lowering supertension by the use of highly dispersed materials as cathodes.

It seems to us that the greatest practical interest is presented by research directed toward a choice of cathode material by means of a systematic study of hydrogen supertension on dispersed electrodes obtained by different methods and from differ-

ent metals and alloys.

In our preceding works we were able to establish [6] the basic conditions for the electrolytic product on of highly dispersed mono and poly-component cathode metal precipitates. The goal of this work was to establish an industrial method of obtaining metals and alloys in powder form for metalloceramics, catalysts and cementing materials in hydroelectrometallurgy.

High purity, dispersion and the high chemical activity of the products obtained gave rise to the assumption that their utilization as cathode material might lead to a sharp increase in the speed of hydrogen discharge. At the same time, the method recommended by us for the electrolytic precipitation of thin layers of dispersed metals and alloys over compact electrodes, may easily be carried out with existing devices and therefore does not entail great expense. In this connection we investigated the kinetics of the separation of hydrogen on electrolytically obtained, dispersed precipitates of Ag, Pt, Cu, Bi, Sn, Mi and Fe.

EXPERIMENTAL PART

Experimental method

The precipitates to be studied were prepared by means of electrolytic separation of the metals on a Pt-electrode. The composition of the electrolytes and the current densities are shown in Table 1. Initially, during the production of dispersed precipitates, a layer of compact metal coated the platinum electrode, then, with current densities surpassing the maximum current density of diffusion,

 $(D_{\rm max})$, the powder precipitated. The value of $D_{\rm max}$ was previously computed according to the equation corresponding to the maximum diffusion speed of the discharged ions, and was then verified experimentally. For the precipitation of bi-component dispersed systems, electrolysis was conducted with current densities higher than the value of $D_{\rm max}$ in order to obtain a more positive metal.

The amount of precipitated powder was the same in all instances. The duration of electrolytic precipitate coating was determined from the relation $D_t = Q\frac{Ak}{t_m^2}$, which equalled 0.008-0.01.

Table 1

CONDITIONS FOR THE PRECIPITATION OF

METALS AND ELECTRICAL CAPACITY

[1]	[2]	[3]	[7+]
Precipitate	Electrolyte Composition	Current Density	Electrical
I TOOLLY TOO		(in mA/cm2)	Capacity
			$(in mF/cm^2)$
Compact Cu	0.1 n. GuSO 1 n. H SO	D ₁ =1.5	35
Powder Cu	Same	D ₁ =1.5; D ₂ =50	1180
Compact Ag	0.005 n. NaAg(CN)	D ₁ =1.5	200
Powder Ag	Same	D ₁ =1.5; D ₂ =50	912
Compact Ni	0.1 n. NiSO 20 g/l (NH)SO	D ₁ =1.5	50
	10 g/l NaCl		
Powder Ni	Same	D ₁ =1.5; D ₂ =50	1025
Compact Pt	Smooth Commercial Platinum		
Powder Pt	Electrolite, Platinated	(*)	1030
	According to Mislovitser		

[1]	[2]	[3]	[71]
Compact Bi	l n. (C H OHCH SO) Bi	D ₁ =10	92
Powder Bi	Same	Dl=2; D2=150	910
Compact Fe	0.1 n. FeSO 20g/1 (MH) SO	$D_T = t$	ent eng
	lo g/l NaCl		
Powder Fe	Same	D ₁ =4; D ₂ =100	1200
Compact Sn	0.2 n. SnSO l n. H SO 0.3	D ₁ =10	35
	g/l diphenylamine 0.75 g/l		
	naphthol		
Powder Sn	0.2 n. SnSO l n. H SO l g/l	D _l =10; D ₂ =100	577
	gelatin		
Powder Fe+Ni	(0.1 n. FeSO 0.02 n. NiSO)	D ₁ =4; D ₂ =100	1,320
	20 g/l(NH) SO lOg/l NaCl		
Powder Cu+Ag	(0.1 n. CuSO 0.02 n. Ag SO)	D ₁ =1; D ₂ =20	1250
	1 n. H SO		
Powder Fe+Cu	(0.1 n. FeSO 0.02 n. CuSO)	D ₁ =1.5; D ₂ =10	1100
	20g/1(NH) SO log/l NaCl		
Powder Sn+Cu	(0.2 n. SnSO 0.20uSO) l n.	D ₁ =1; D ₂ =50	950
	H SO		

^(*) Current from a 4-volt Accumulator was used for five minutes

Table 2

YIELD
POTENTIALS OF HYDREOGEN SEPARATION ON COMPACT AND DISPERSED METALS (*)

г	1]						[2]							[3]	
						Electr	olyte in S	ulphuric	Acid					Electro	lyte
1.	D A/cm ²						·			-				22 percen	t KOH
m.e	t/ cm-	,	[A]	[]	31	[c]		[D]		[E]		[F]		[G]	
			Cu		Ag	Ni	Ĺ	Pt	;	Bi		Sn		Fe	
		[a]	Сц [b]	[c]		[e]	[f]	[g]	[h]	[i]	[j]	[k]	[1]	[m]	[n]
		Smooth	Powder	Smooth	Powder	Smooth	Powder	Smooth	Powder	Smooth	Powder	Smooth	Powder	Smooth	Powder
i Vi I		200 p F	1180 p F	200 p F	912 µF	50 µF	1025 pt		1030 µ F	92 pF	910 µF	35 µ F	577 µF		1200 <i>p</i> ·F
	0.25	0.350	·	0.342	0.232	0.313	0.377		0.283			0.534	0.537	1.292	1.192
	0.50	0.479	0.453	0.523	0.380	0.356	0.378			0.800	0.620	0.550	0.541	1.336	1.252
	1.00	0.545	0.490	0.558	0.406	0.410	0.379	0.557	0.290	0.860	0.745	0.830	0.578	1.394	1.278
	2.00	0.590	0.531	0.580	0.433	0.473	0.381	0.569	0.296	0.895	0.782	0.874	0.780	1.462	1.303
	4.00	0.643	0.565	0.621	0.448	0.526	0.389	0.588	0.300	0.932	0.813	0.910	0.818	1.522	1.332
	6.00		0.585	0.639	0.459	0.562	0.397	0.625	0.302	0.952	0.834	0.933	0.840	1.567	1.348
	8.00	0.682	0.585	0.657	0.466	0.583	0.413	0.656	0.304	0.962	0.848	0.949	0.855	1.593	1.357
	10.00		0.590	0.676	0.470	0.600	0.418	0.680	0.306	0.970	0.859	0.960	0.859	1.614	1.366
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[1]						. [2]						[3	3]
			[A]	[B]	[C]	[D]	[E]		[F]		[G]	
		[a]	[b]	[c]	[d]	[e]	[f]	[g]	[h]	[i]	[j]	[k]	[1]	[m]	[n]
5	0.0	0.870	0.665	0.870	0.520	0.817	0.496	0.817	0.325	1.032	0.926	1.091	0.921	1.786	1.418
6	0.0	0.860	0.662	0.889	0.532	0.847	0.504	0.834	0.328	1.040	0.934	1.126	0.930	1.801	1.428
7	0.0	0.880	0.670	0.903	0.532	0.870	0.510	0.849	0.332	1.047	0.940	1.154	0.936	1.816	1.438
8	0.0	0.880	0.672	0.926	0.535	0.922	0.511	0.863	0.334	1.052	0.945	1.180	0.941	1.834	1.445
9	0.0	0.890	0.675	0.938	0.541	0.948	0.516	0.878	0.337	1.057	0.951	1.197	0.947	1.847	1.452
10	0.0	0.900	0.679	0.950	0.544	0.974	0.522	0.892	0.340	1.060	0.960	1.220	0.950	1.863	1.458

^(*) Here and in Table 3 the values of E are given in relation to the potential of a normal calomel electrode (E=0.28) taken as zero.

														[3]		
	F2.1						[2]	l		rm)		[F]		[G]		
	[1]	[1	1]	[B]		[C]	•	[D]		(E)	[j]	[k]	[1]	[m]	[n]	
		[a]	[b]	[c]	[d]	[e]	[f]	[g]	[h]	[i]	[3]			3 (27	1.372	
					0 : 21	0.623	0.428	0.699	0.307	0.978	0.866	0.964	0.867	1.637		
	12.00	0.701	0.596	0.682	0.474		0.439	0.714	0.308	0.983	0.870	0.972	0.872	1.655	1.378	
	14.0	0.711	0.601	0.690	0.478	0.640			0.311	0.987	0.877	0.984	0.877	1.671	1.382	
	16.0	0.723	0.610	0.706	0.481		0.447	0.730		0.990	0.880	0.992	0.881	1.681	1.385	
	18.0	0.733	0.614	0.730	0.484	0.649	0.451	0.740	0.312		0.883	0.999	0.855	1.691	1.388	
		0.743	0.620	0.758	0.486	0.670	0.456	0.749	0.313	0.993		1.007	0.889	1.699	1.391	
	20.0		0.624	0.780	0.489	0.686	0.461	0.757	0.313	0.997	0.885			1.706	1.392	
1	22.0	0.753			0.490	0.700	0.466	0.765	0.313	1.00	0.890	1.018	0.891		1.394	
0	24.0	0.770	0.628	0.800			0.469	0.772	0.314	1.003	0.894	1.029	0.894	1.716	_	
	26.0	0.780	0.632	0.820	0.492	0.715		0.778	0.314	1.005	0.897	1.038	0.896	1.721	1.395	
	28.0	0.780	0.634	0.825	0.493	0.720	0.471		0.314	1.008	0.900	1.046	0.900	1.731	1.395	
	30.0	0.805	0.636	0.831	0.495	0.743	0.475	0.784			0.913	1.066	0.912	1.765	1.408	
	40.0		0.647	0.855	0.505	0.784	0.487	0.798	0.322	1.024	0.71)		•		a T	

FIGURE 1

Device for the Determination of Polarization.

- (1) Hydraulic seal. (2) Cut. (3) Haber.
 - (4) Cathode. (5) Anodes.

The measurements were made in an electrolyzer, snown in Figure 1, made of a glass vessel containing two Pt-anodes and of an electrolytically coated Pt-Cathode with S=3 cm². The hydrogen separation potential was measured in an 1 n. solution of $\rm H_2SO_4$, or in a 22 percent solution of KOH (chemically pure). The ends of the haber came to both sides of the cathode and were pressed tightly against the metal layer. The haber, through intermediate vessels was connected with a normal calomel electrode which serve as comparison electrode.

The measurements of the potential (\mathbb{F}_k) were made by the direct compensation method with the help of a Raps type potentiometer built at the Etalon plant. The electrolyzer was put in a thermostat in which a constant temperature of 30±0.5 degrees centigrade was maintained by an electronic relay.

bottom. In order to insure a constant hydrogen atmosphere in the instrument, the latter communicated with the outside atmosphere only by way of a hydraulic seal. The supertension measurements were made during a strictly constant number of stirring rod revolutions. This was controlled by a voltmeter connected with the motor cleats.

Together with the supertension on measurements we determined the electrolytic capacity of the cathode in proportion to its actual surface. The capacity was measured upon polarization of the electrode

-- 0.6-0.8 V. The method for capacity measurement was described earlier.

DISCUSSION OF RESULTS

The research data concerning hydrogen separation potentials in 1 n. $\rm H_2SO_{\mbox{${\bf l}$}_1}$ and 22 percent KOH solutions on compact metals: Pt, Bi, Ni , Ag , Sn , Cu , Fe and their dispersed pricipitates, as well as on their alloys, is given in Tables 2 and 3 and in Figures 2 and 3. In all cases the values or the potentials are given in relation to the potential of a normal calomel electrode. In order to convert these figures to the hydrogen scale one must subtract 0.28 V from the Table values.

Table 3 POTENTIALS OF HYDROGEN SEPARATION ON DISPERSED BI-COMPONENT PRECIPITATES OF METALS.

		PRODUPLIE	1100 01 1000			0
ELECTROLY	TE 1 n.	H2SO4 ANI) 22PERUENT	KOH. TEM	ipdrature,	30°
[1]	[2]	*	[3]		[4]	[5]
D	Electro	lyte	Electroly	rte		
mA/cm^2	in H ₂ S	10 ₁	22 percent	t KOH	I mA	lg I
1127	[a]	[b]	[c]	[d]		
	1250 p.F	950 µF	1100 pc F	1320µF		
0.25	and the	0.510	1.204	1.214	0.75	-0.125
0.50		0.530	1.276	1.246	1.50	+0.176
1.00	0.333	0.724	1.300	1.287	3.00	0.477
	0.378	0.780	1.320	1.310	6.00	0.778
2.00	0.510	0.817	1.343	1.332	12.0	1.079

0.817

0.402

4.00

[1]	[2]		[3]		[4]	[5]
	[a]	[b]	[c]	[d]		
6.00	0.414	0.834	1.360	1.344	13.0	1.255
8.00	0.423	0.843	1.374	1.35	24.0	1.380
10.00	0.431	0.852	1.383	1.362	30.0	1.477
12.0	0.1437	0.858	1.393	1.369	36.0	1.556
14.0	0.442	0.865	1.400	1.375	42.0	1.623
16.0	0.14248	0.872	1.405	1.380	48.0	1.681
18.0	0.1452	0.876	1.410	1.385	54.0	1.732
20.0	0.457	0.880	1.416	1.390	60.0	1.778
22.0	0.461	0.884	1.420	1.394	66.0	1.819
24.0	0.467	0.888	1.425	1.397	72.0	1.857
26.0	0.471	0.892	1.429	1.401	78.0	1.892
28.0	0.478	0.894	1.433	1.406	84.0	1.924
30.0	0.486	0.896	1.436	1.410	90.0	1.954
40.0	0.506	0.905	1.455	1.424	120.0	2.079
50.0	0.521	0.913	1.470	1.436	150.0	2.176
60.0	0.535	0.920	1.480	1.447	180.0	2,255
70.0	0.545	0.928	1.489	1.457	210.0	2.322
80.0	0.550	0.937	1.496	1.466	240.0	2.380
90.0	0.555	0.944	1.503	1.473	270.0	2.431
100.0	0.560	0.950	1.508	1.480	300.0	2.477

Examining the data in the tables we see that in all cases the hydrogen separation potential on dispersed metals and their alloys is considerably lower than on compact metals. Thus, for instance, the following values, obtained from Table 2, show that the difference between the hydrogen separation potentials on compact and dispersed silver,

when $D=100\text{mA/cm}^2$, is of 100mV, on nickel cathodes 152mV, etc..

Electrode Material	Pt	Fe	Cu	Ag	ľ.ί.	Łd	Sn
$\Delta E_{v}^{=E}$ comp $^{-E}$ disp when D=100mA/cm 2	0.552	0.405	0.221	0.406	0.452	0.100	0.270
Increase in surface S _{disp} : S _{comp}		 20	35	4.5	20	10	16

Especially intersting is the fact that the drop in polarization during passage to dispersed precipitates is almost in every instance significantly bigger than would be true if only the drop in the current density were considered. In reality, let us suppose, that the relation of $\;\eta\;$ to D, in accordance with the retarded η= α +0.116 lg D. discharge theory, is subject to the equation:

Then, for the mickel cathode, for instance, the difference of ${\rm H_2}$ supertension on dispersed and compact electrodes must equal:

while the actual difference is γ_c - γ_d =0.452V, or approximately three times more.

FIGURE 2

 $\rm H_2$ Separation Potentials on Smooth and Dispersed Metals in 1 n. $\rm H_2SO_{\underline{l}_1}$ 22 percent KOH (Fe).

(A) D mA/cm²; Abscissa Axis -- E_b . Metals: (1) Pt_n; (2) N_{in} ; (3) Ag_{n} ; (4) Cu_{n} + Ag_{n} ; (5) Cu_{n} ; (6) Pt_{2} ; (7) Cu_{2} ; (8) Ag_{2} ; (9) Ni_{2} ; (10) Sn_{n} + ${\rm Cu}_{\rm n};$ (11) ${\rm Sn}_{\rm n};$ (12) ${\rm Bi}_{\rm n};$ (13) ${\rm Bi}_{\rm 2};$ (14) ${\rm Sn}_{\rm 2};$ (15) ${\rm Fe}_{\rm n};$ (16) ${\rm Fe}_{\rm n}+{\rm Ni}_{\rm n};$ (18) Fe2.

A similar regularity is observed for Ag, Pt, Sn, and other metals. Thus a change from compact to dispersed cathodes leads to a drop in supertension not only because of an increase in the actual electrode surface, but, in a number of cases, principally because of an increase in the discharge speed constant. The latter is in good accord with the heightened chunical activity of a series of electrolytic powders in purely chemical reactions.

When a certain quantity of a more electropositive metal enters into the dispersed precipitate (bi-component system), the drop in supertension is still more significant. For instance, in changing from compact to dispersed copper, the drop of the hydrogen separation potential is:

$$E_{comp}$$
. Cu — E_{powd} . Cu+Ag=900—560=340mV.

The values of the potential correspond to the straight lines in the semi-log rithmic system of coordinates E = lgI, and the values of the coefficient b in Tafler's formula ($\eta = a - llgD$) for a series of smooth metals are close to the data of Pecherskaya and Stender, giving an average of 0.10-0.12. The values of the coefficient b for smooth (compact) Ni, Fe, Cu have been found to be somewhat higher (Table 4).

See page 13 for Table 4

for the majority of dispersed metals, the value of the coefficient b is less than 0.10. This agrees well with the data of various authors who think that like retarded discharge supertension may largely be determined by other "stages" of hydrojen separation, in particular, by the retarded state of the process of recombination of the hydrogen atoms.

Table 14

VALUED OF THE COMPTICIENT B FOR THE HYDRODEN SUPERTENSION

ON DISPERSED ELECTROLYTIC PRECIPITATES

Electrode Material	Coefficient (b)	Electrode	Material	Coefficient (b)
Pt	0.015		Pe+Cu	0.111
Fe	0.087		Fe+Ni	0.075
Ag	0.060		Cu+Ag	0.082
B1	0.096		Sn+Cu	0.0945
Sn	0.091			
Cu	0.150			

The experimental material here presented shows that the employment of electrolytic coating of cathodes by dispersed metals and their alloys opens prospects for a singificant drop (in certain cases up to 0.4V and more) in cathode potential in industrial processes, for example during the electrolytic production of chlorine, electrolysis of water, and others. This fact will considerably lower the consumption of electric power and, consequently, will reduce the cost of the processes.

FIGURE 3

Relation E-lgI for H_2 Separation. The value $E_{equilib}$ is given for the separation of H_2 from 1 n. H_2SO_4 (0.29V in relation to a calomel electrode). B -- I(in mA); A -- E(in mV); V -- lgI; G -- $E_{equilib}$ Metals: 1 -- $F_{e}+Cu; 2$ -- $F_{e}; 3$ -- $E_{i}; 4$ -- $E_{i}; 5$ -- E_{e} Sn+Cu; 6 -- E_{e} Ag; 7 -- E_{e} Cu; 9 -- E_{e} Ag; 10 -- E_{e} Pt; 11 -- E_{e} Mi.

CONCLUSTONS

- (1) We studied the supertension of hydrogen in 1 n. solution of $H_2SO_{\downarrow\downarrow}$ and 22 percent KOH on compact and dispersed electrolytic precipitates of Pt, Ag, Cu, ti, Sn, Ni and Fe and also on bi-component metallic systems (Cu-Ag; Sn-Cu; Fe-Ni).
- (2) We showed the possibility of lowering the supertension by more than 0.4V by using certain powdery electrolytically obtained, electrodes.
- (3) We established that the drop of supertension occurs not only because of the lowering of the actual current density in relation to the increase in the actual surface of the cathode, but also in great measure because of the growth in the value of the discharge speed constant of hydrogen ions.

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